

partitioning the surface into regions formed from different thiols¹²⁻¹⁵. Here we describe a way to pattern SAMs using a single alkanethiol on substrates consisting of regions of different topography: planar islands of one metal on the surface of a second (which may be different from or the same as the first). These topographically patterned SAMs consist of three regions: two planar surfaces and a transition region between the two. The characters of the SAMs on these three regions were inferred from images of three structures that form on them: condensation figures, patterns of crystals of CaCO₃ and regions of selective etching. The transition region is more active in the processes generating these structures than either of the two planar regions, and we propose that this activity is due to the relatively high disorder in the organic film there. We believe that this ability to

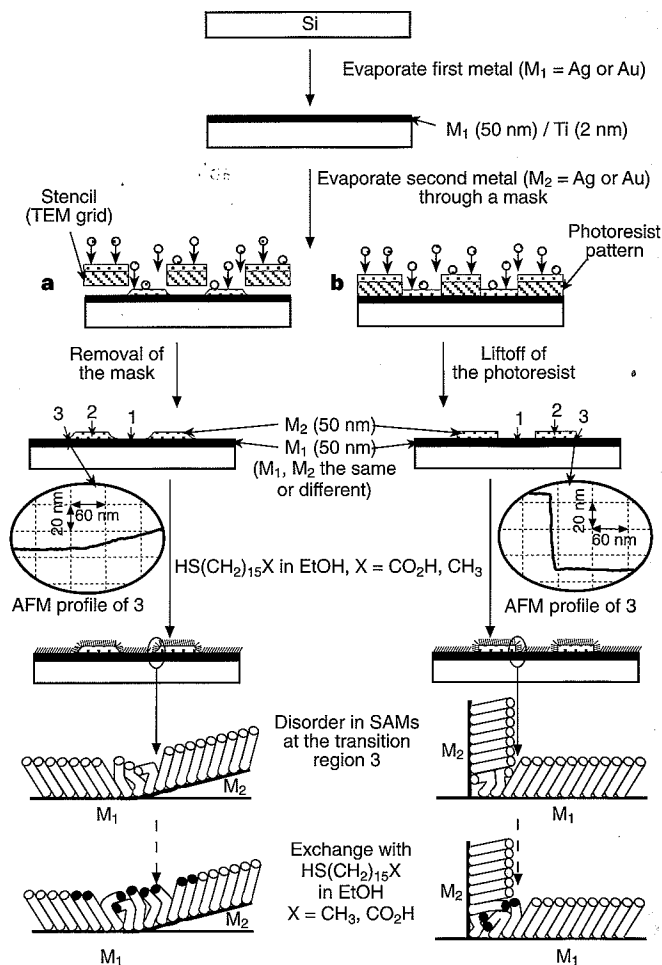


Figure 1 Schematic representation of the procedures used for fabrication of SAMs supported on topographically patterned metal surfaces. **a**, Patterned deposition of the overlayer of a metal on the surface of another metal (which might be the same or different) through a stencil. **b**, Area-selective deposition of the overlayer of a metal on the surface of another metal protected by a pattern in photoresist, followed by lift-off of the photoresist. The height profiles of the micropatterned surfaces recorded using atomic force microscopy (AFM) show that procedure **b** generates sharper edges for the patterned features than procedure **a**. The details of the surface in the transition region at the level of the grain and domain sizes and boundaries were not studied. The structure of the SAM shown is schematic. In the experiments involving formation of condensation figures, we allowed the thiol in the transition region of the SAM to exchange with a different thiol in solution. To form the patterns, silicon wafers (test grade, n or p type; Silicon Sense, Nashua, NH) were coated with 2.5 nm of Ti to promote adhesion, and then with 50 nm of metal (Ag or Au) using an electron beam evaporator and a stencil mask or patterned photoresist.

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Controlling local disorder in self-assembled monolayers by patterning the topography of their metallic supports

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Micropatterning is a powerful method for controlling surface properties, with applications from cell biology to electronics¹⁻⁸. Self-assembled monolayers (SAMs) of alkanethiolates on gold and silver⁹⁻¹¹—the structures most widely used for preparing organic films with specific surface properties—are usually patterned by

control the local disorder in a SAM with high resolution will be important in controlling processes such as nucleation, wetting, adhesion and etching on scales of below 50 nm to 5 μm .

In this approach to micropatterning, we fabricate patterned substrates by evaporating one metal (Au or Ag) onto the surface of a second (Au or Ag) through a stencil mask or a patterned photoresist (Fig. 1). We describe these surfaces using the nomenclature M_2 -on- M_1 ; that is, M_2 (the second metal) evaporated on M_1 (the first). We formed SAMs on these patterned substrates by immersing them in a 10 mM solution of $\text{HS}(\text{CH}_2)_{15}\text{X}$ ($\text{X} = \text{CH}_3$ or CO_2H) in ethanol for 1 h. SAMs of the same alkanethiol formed on different metals differ in structure⁹⁻¹¹, and we expected, therefore, that the SAM at the border between two different metals would have a distinct and probably less-ordered structure (region 3 in Fig. 1) than the larger planar regions (regions 1 and 2 in Fig. 1). It seemed possible that the region of SAM at sharp steps in the topography of the same metal might also have abnormal structure.

To test the hypothesis that the transition region would be disordered, the topographically patterned substrates supporting one thiolate—either methyl- or acid-terminated—were allowed to soak in a solution of the second (10 mM). We expected the disordered regions of SAMs to be more labile than the ordered regions, and thus to exchange selectively¹⁴⁻¹⁶. We examined the wettability of the surfaces using condensation figures¹⁷⁻¹⁹ to visualize modifications in the SAM resulting from this exchange (Fig. 2). The best visualization of the transition region was achieved when we used long exchange (~ 10 h) with the alternative thiol. When a methyl-terminated SAM was allowed to exchange with the acid-terminated thiol, condensation figures on the modified surface consisted of elongated drops of water at a hydrophilic border between the two metals (Fig. 2a-c). When the SAMs were formed in the reverse order—first from the acid-terminated thiol and then

exchanged with methyl-terminated thiol—the pattern of wetting was also reversed: water drops nucleated everywhere except at a hydrophobic border between the metals (Fig. 2d, e). The condensation figures outlining transition regions were particularly pronounced when M_1 was not the same as M_2 (Fig. 2b, c, e). The wetting of surfaces that were not subjected to exchange was largely homogeneous across the surface (Fig. 2a-e), with insignificant preference for nucleation at the boundaries when M_1 was not the same as M_2 presumably due to the pinning of water drops by the local disorder in the SAM. Exchange experiments using substrates prepared by deposition of M_2 at an angle of $\sim 20^\circ$ from the normal show that the size and shape of the exchanged region—which is related, we presume, to the structure of the disordered SAM—depend on the topology of the boundary between the two metals (Fig. 2c).

These preferential exchange experiments establish that there are three distinct regions in the SAMs supported on micropatterned mixed metal substrates, and suggest that one of these regions is a narrow region of a labile, disordered SAM at the border between the two metals. So this technique provides, to our knowledge, a new concept in patterning surfaces—controlling the local disorder in SAMs on a submicrometre scale by manipulating the topography and composition of the supporting metals. The boundaries between the two metals present disordered regions on the surface, and can be thus used to pattern processes that are sensitive to surface structure. For processes that require the formation of narrow features, the best system is a substrate in which M_1 is the same as M_2 , patterned using a photoresist mask to produce vertical walls in the relief. For processes that do not require the formation of narrow features, a combination of different metals gives a more pronounced effect. We demonstrate two further examples of this method: nucleation of crystal growth, and etching of the underlying metal.

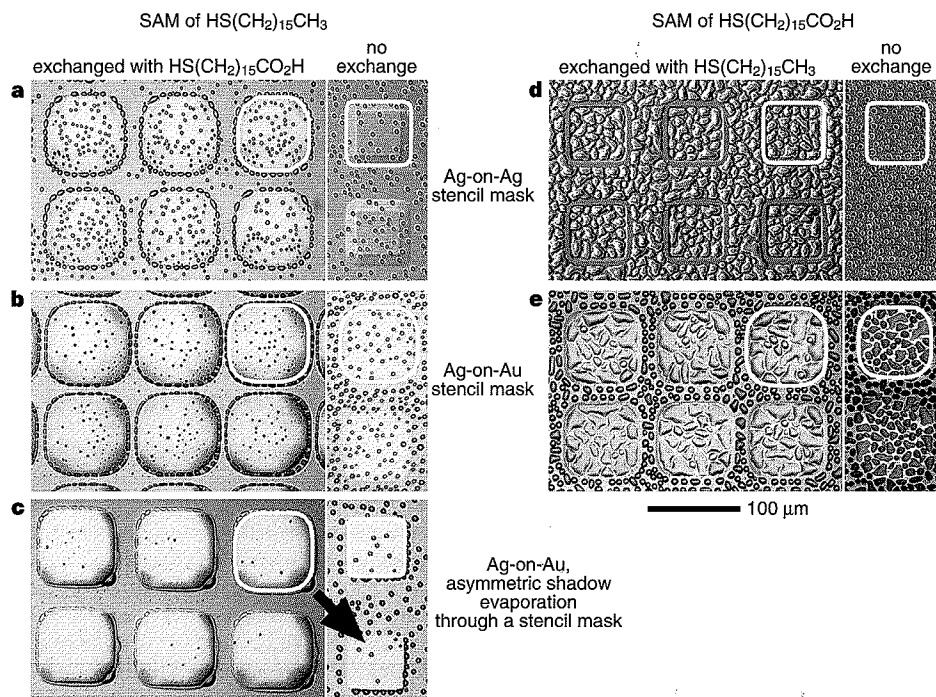


Figure 2 Optical micrographs of the condensation figures (CFs) formed on SAMs of alkanethiols supported on micropatterned metal substrates. These substrates were fabricated by depositing silver (50 nm) on silver (a and d), or silver (50 nm) on gold (b, e and c) using a stencil (a TEM grid). The geometries of the islands of the second metal (shown in overlays) depend on the distance between the mask and the substrate and on the angle between the beam and substrate. a-c, CFs formed on SAMs of $\text{HS}(\text{CH}_2)_{15}\text{CH}_3$ exchanged with $\text{HS}(\text{CH}_2)_{15}\text{CO}_2\text{H}$; d, e, CFs formed on SAMs of $\text{HS}(\text{CH}_2)_{15}\text{CO}_2\text{H}$ exchanged with $\text{HS}(\text{CH}_2)_{15}\text{CH}_3$. The rightmost columns

of each pattern show CFs formed on the control patterned substrates supporting corresponding SAMs that were not subjected to exchange. The surfaces were derivatized initially by exposing them to a solution of one thiol (10 mM in ethanol) for 30 s. The substrates were rinsed with ethanol and allowed to soak in a solution of the second thiol for 10 h. We observed CFs directly by breathing on the samples under an optical microscope. The substrate in c was prepared by asymmetric shadow evaporation of Ag through a stencil mask—the direction of the beam (indicated by an arrow) was $\sim 20^\circ$ from the normal to the surface.

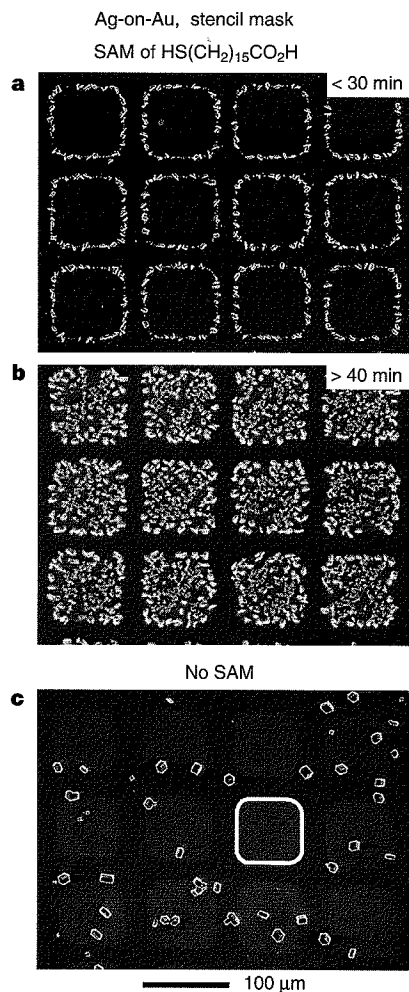


Figure 3 Scanning electron micrographs of calcite crystals grown on topographically patterned surfaces fabricated by depositing Ag (50 nm) on Au through a stencil mask followed by formation of SAMs. The overlay shows the outline of the pattern. **a**, **b**, Patterns of calcite crystals formed on SAMs of HS(CH₂)₁₅CO₂H supported on micropatterned metal surfaces. For short times of crystallization (<30 min), calcitic outlines of the underlying patterns are formed (**a**). For longer times of crystallization (>40 min), preferential filling of the Ag regions with crystals takes place (**b**). **c**, Non-patterned growth of calcite induced by bare substrates supporting no SAM. The patterned metal surfaces derivatized with a SAM of HS(CH₂)₁₅CO₂H were used directly in the crystallization experiments (no exchange was performed). The substrates were placed with the pattern down into a 25 mM CaCl₂ solution in a closed desiccator containing vials of solid (NH₄)₂CO₃.

Crystallization is highly sensitive to the structure and order of the surfaces that induce nucleation^{20–22}. We used the formation of calcium carbonate as a model system^{22–24}. The micropatterned Au-on-Ag and Ag-on-Au substrates supporting a CO₂H-terminated SAM induce patterned crystallization (Fig. 3). The crystals nucleate first at the interface between the two metals (Fig. 3a), and then fill the Ag region (Fig. 3b). The bare, micropatterned metal surfaces (with no SAM) do not induce preferential nucleation anywhere on the surface (Fig. 3c); this observation indicates that patterned nucleation arises from differences in the structure of the patterned SAM and not directly from the topography of the surface. Figure 3 suggests that these techniques may provide a simple method to pattern crystal growth²⁵. The fabrication of crystalline outlines (Fig. 3a) is especially striking, as these structures cannot be produced using other methods of crystallization.

The hypothesis that the transition regions in the SAMs are

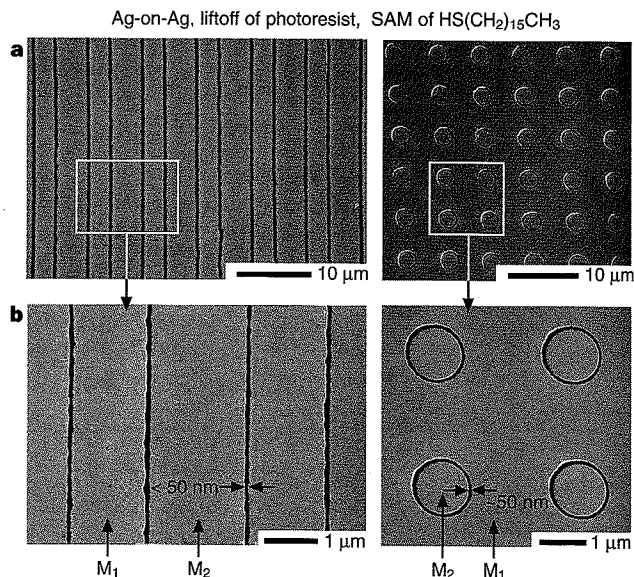


Figure 4 Selective etching of micropatterned substrates fabricated by depositing Ag (50 nm) on a Ag film through a layer of patterned photoresist, followed by lift-off. Left panels, 2- μ m-wide lines; right panels, a square array of circles with 1.5 μ m diameter. **a**, Low magnification SEM, showing the homogeneity of the pattern. **b**, Trenches (\sim 50 nm wide) in the metal substrates fabricated by etching for 10 s. Substrates were exposed to a HS(CH₂)₁₅CH₃ solution (10 mM in ethanol) for 1 h and rinsed with ethanol, then etched with an aqueous solution of ferrocyanide (0.1 M), ferricyanide (0.001 M) and sodium thiosulphate (0.1 M).

disordered suggests that these regions expose the underlying metal to etching^{15,26}. When Ag-on-Ag or Au-on-Ag surfaces supporting a SAM of HS(CH₂)₁₅CH₃ were etched with an aqueous solution of potassium ferricyanide, dissolution of silver initiated preferentially at the border between the two metals (Fig. 4a). In control experiments with bare, topographically patterned substrates supporting no SAM, corrosion was uniform across the surface, and was not selective at the edges of the metal patterns. We suggest, again, that the specificity of etching at the transition regions results primarily from the disorder of the SAMs. The ability to etch the disordered regions in the SAM selectively provides a method to form small features in metal films^{1–3}. The structures generated by etching topographically patterned surfaces supporting SAMs are smaller than the original patterns used for their generation, and their feature sizes can be controlled by varying the etching time. Figure 4b shows trenches \sim 50 nm wide in the metal substrates fabricated by short (10-s) etching, using two sample patterns of lines (Fig. 4b, left) and circles (Fig. 4b, right).

The ability to control the local disorder in SAMs at the nanometre scale therefore provides an approach to high-resolution patterning of surfaces, and constitutes a convenient method for size reduction: features less than 50 nm wide can be generated at the edges of micrometre-sized patterns. □

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